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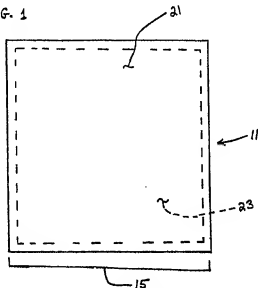
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(57) Molten metal filter medium and method for making same.

(57) A porous ceramic body of high structural strength and integrity is disclosed, along with the method of fabricating such, whereby a curable resin and a sinterable ceramic are mixed and then admixed to removable pore formers, then consolidated into a green body, the pore formers removed, and sintered into the porous ceramic body.

FIG. 1



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MOLTEN METAL FILTER MEDIUM AND METHOD FOR MAKING SAME

## BACKGROUND OF THE INVENTION

## 1. Field of the Invention

This invention relates generally to the field of filtering contaminants from molten, or liquid phase metals and particularly to a means for separating non-metallic inclusions and contaminants from molten metal as it is flowed.

## 2. Background

In the melting, refining and forming of metals, typically when molten metals are cast, it is desirable to separate exogenous intermetallic inclusions from the molten metal. Such inclusions result, in molten metals, from impurities included in the raw materials used to form the melt, from slag, dross and oxides which form on the surface of the melt, and from small fragments of the refractory materials that are used to form the chamber or vessel in which the molten metal melt is formed. Such inclusions, if not removed from the molten state of the metal, can result in weakened points and/or porosity in the final formed and solidified metal body which is the eventual downstream end product of the melting operation.

Typically, in a metal casting operation, the metal melt is formed in a furnace wherein the constituent components are added in the form of unmelted scrap and/or refined virgin metal, deoxidizing agents in various forms (both solid and gaseous or a combination of both) and alloying elements. Very light (less dense) solids and gases tend to migrate to the surface of the melt where they either effervesce or float in combination with partially and completely solidified oxides known variously as slag and dross. The higher density impurities in the melt tend to remain in some degree of suspension in the liquid phase of the metal, or melt, as the fluid flow convection currents are generated within that melt by the heating means applied by the melting furnace.

In the melting operation, the furnace acts as a vessel to hold the molten metal as it is being melted and, depending on the particular species of molten metal or alloy being formed, for a period of time following melting, to refine the molten metal by way of the gases and low density impurities migrating to the surface. The molten metal is then transferred, typically to another vessel, such as, for example, a ladle, to transport it to the forming means, such as, for example, a mold. Alternatively, the molten metal may be drawn directly from the furnace and flowed by gravity through a channeling means to a forming means, such as a continuous caster. A variety of other methods, used for removing molten metal from a furnace and conveying it to a forming means, are well known to those with skill in the art.

During this transportation or conveyance phase, wherein the molten metal is moved from the melting furnace to the forming means, it is desirable to ensure that the dross or slag, from the surface of the melt, does not become included in the formed metal and, also, that the higher density, exogenous intermetallic inclusions in the melt are not included in that formed metal.

One method that is used to prevent the inclusion of exogenous intermetallic substances, including slag or dross, in the formed metal body is to filter the molten metal as it is flowed from the melting furnace to the forming means. A variety of means for accomplishing this filtration are well known to those with skill in the art. Recent examples of this can be found in U.S. Patent Nos. 4,444,377; 4,426,287; 4,413,813; 4,384,888; 4,330,328; 4,330,327; 4,302,502; 4,298,187; 4,258,099; 4,257,810; 4,179,102; 4,159,104; 4,081,371; 4,032,124 and 3,869,282. There are many additional recent references readily available that demonstrate methods of filtering molten metal. Also, there are many older references available, which date further back into history and which show apparatus and methods for filtering molten metals, such as, for example, U.S. Patent No. 3,006,473.

In such systems, a filter medium or filter element is used. The basic material property that is necessary in a filter medium is that it be formed from a high temperature material which will withstand the elevated temperatures of molten metals and be stable in such an environment. That is to say that the material must not be subject to deterioration from melting, chemical reactions or erosion at such elevated temperatures. Also, the filter medium must maintain structural integrity at such elevated temperatures. And, of course, to act as a filter, the filter medium must be capable of either entrapping or preventing the flow of solids, liquids, and semi-liquids, all of which are non-metallic or intermetallic, either by chemically reacting the filter

medium material with such inclusions and/or by mechanically preventing the flow thereof through the filter medium, while still permitting and facilitating the flow of the molten (liquid) metal therethrough. Further, such filter media are used in production facilities in association with unskilled or semi-skilled labor and heavy industrial machinery, equipment and tooling. Thus, such filter media should exhibit a high degree of structural integrity, at room temperature, such that rough handling will not be detrimental.

Many different designs of filter medium are known to those with skill in the field. Also well known are the uses of many different materials for, and many different methods of fabricating, or producing, porous bodies which can be used as filters. U.S. Patent No. 3,796,857, for example, teaches the use of a fluidized and sintered aggregate of particles to form a porous chromatographic filter medium for separating gases from liquids and different liquids from each other. U.S. Patent No. 4,430,294, as another example, teaches the formation of porous nickel bodies by using reducing gases and carbon powder to form interstices in nickel powder, during a rapid sintering process. U.S. Patent No. 4,285,828, as yet another example, teaches forming a porous aluminum body by combining aluminum powder with an expanding agent, such as a fine salt, hot pressing the mixture and dissolving the expanding agent from the pores of the body. U.S. Patent No. 4,391,918, as yet another example, shows the impregnation of an open celled organic foam with a slurry composed mostly of aluminum oxide plus sintering aids. The organic foam is then burned out as the slurry is sintered to form an aluminum oxide ceramic foam which can be used to filter molten metals.

Many older patents teach the bonding of crystalline ceramic material, such as silicon carbide or alumina, with a vitrified ceramic material such as glass. Such is taught, for example, by U.S. Patent No. 2,007,053. Also, it is known to directly sinter particles of ceramic material into a porous body to form a filter medium. Such is taught, for example, by U.S. Patent No. 2,021,520. Finally, it is known to mix a ceramic material such as aluminum oxide with a combustible material such as carbon and burn out the combustible material during sintering to produce a porous body. Such is shown, for example, in U.S. Patent Nos. 2,360,929 and 2,752,258.

One of the problems that is inherent in many of the filter media which are useful for filtering molten metals is that it is difficult to render the pores or passageways throughout the filter media substantially open but also controlled in sizing such that the molten metal will freely flow through the filter media cross section at a controlled rate and so that all solid matter, of a calculated size range or larger, will be uniformly blocked from passage through the full cross section of thickness of that filter media.

Another problem that is inherent in many of the known filter media which are useful for filtering molten metals is that the surfaces of the pores or passageways through the filter media are not smooth, and thus, are susceptible to non-uniform build up of solid matter which tends to adhere more readily to the non-smooth surfaces adjacent to the entry side of the filter medium, thus not fully utilizing the thickness of that filter medium to fully trap such solids. Further, non-smooth surfaces tend to create turbulence in the flow of molten metal, thus inhibiting the smooth flow thereof. These phenomena shorten the useful life of the filter medium as the flow of molten metal therethrough decreases at a relatively greater rate than if the full thickness of the filter media were usable to trap the solid matter.

Another problem that is inherent in many of the known filter media is that it is difficult to localize sizes of pores or passageways through the filter media to form pore size gradations either through the filter media or from one side to another across the face thereof, when desired. Such gradations are useful in specialized situations for preventing the passage of mixed solid materials of various types, and enhancing the separation of certain gases from the molten metal, as it flows through the filter media. Such gradations may also be used to selectively control flow rates in specialized circumstances.

Yet another problem that is inherent in many of the known filter media is that they are too brittle or too friable, or both, at elevated temperatures as well as at room temperature. Thus structural failure of the known filter media has been a major problem related to the economics of using such for filtering molten metal. The strength is known to be diminished by the presence of sharp corners, non-continuous ceramic structure, and large pores in the load bearing sections of the ceramic material. For example, filamentary pores, left behind after the formation of reticulated foam filter media, exhibit such defects.

The present invention provides a filter medium, and a means for producing it, without filamentary pores, but with relatively uniform cell sizes, with passageways or pores therebetween, with relatively smooth surfaces on the cell walls and the walls of the passageways or pores therebetween, with the edges, or discrete transition areas, between cells and their interconnecting pores, being rounded off or smoothed, and a means for forming a filter medium with cell size gradations or localization of cell sizes or passageway (pore) sizes. Also, the present invention provides a filter medium with a high degree of structural integrity both at room temperatures and at the elevated temperatures associated with molten metals.

SUMMARY OF THE INVENTION

The present invention includes a method of making a porous ceramic body as well as the porous ceramic body itself. A quantity of bondable or sinterable ceramic powder, or mixtures of different ceramic powders which will either bond or sinter together, are thoroughly mixed with a polymer binder. A pore former is also mixed with the powder or powders and a polymer binder. The resultant mixture of the constituent components is then consolidated into a relatively dense self-sustaining body. The pore former can be any suitable solid that can be removed by leaching, melting or pyrolysis. The pore former may include therewith a lubricant which is not soluble in the polymer binder and which does not have deleterious effects on the curing of the polymer binder. The pore former, on the other hand, may have inherent self-lubricating properties. Or, alternatively, the resin may by itself, or in combination with other materials, provide lubricity between the pore former surfaces and the resin. The pore former may, also, be deformable under pressing conditions. The polymer binder may be any thermosetting or thermoplastic resin that can be readily mixed with ceramic powder or powders and the pore former, provided that the polymer binder does not dissolve or dilute the lubricant, used in association with the pore former, therein. The self-sustaining body is then consolidated to a shaped size, for example, pressure compaction by die pressing. The polymer binder is then cured. Then the pore former is removed, and the shaped body is elevated to sintering temperature to form a porous sintered ceramic body with cells and interconnecting pores therebetween, in a form which is known in the field of geology as "moldic porosity". The sintered ceramic body has smooth walled cells formed by the pore formers, and pores, with rounded edges, which interconnect the cells, the rounded edges forming the transition points between the cells and the pores. The cell and pore characteristics may be controlled or affected by the deformability, size, distribution and location of the pore formers, and the type, distribution and amount of lubricant means used, and also by the type of consolidation process used, the composition of the sinterable or bondable ceramic powder, or mixture of those ceramic powders, and the type of polymer binder which is used. Either by itself, or in combination with a lubricant, the selected polymer binder should be such that it tends to bead in contact with the surface of the pore former. The ceramic powders, as sintered to form the ceramic body, form a continuous, uninterrupted, void-free and dense sintered ceramic matrix interspersed between the interconnecting voids and pores, which exhibits low friability and low brittleness in regard to physical shock. The porous ceramic body of the present invention is useful for molten metal filtration as well as a variety of other applications, such as, for examples, catalyst supports and gas dispersion mechanisms.

BRIEF DESCRIPTION OF THE DRAWINGS

FIG. 1 is a schematic plan view of a molten metal filter formed from a porous ceramic body according to a preferred embodiment of the shape of the present invention.

FIG 2 is a schematic elevational view of the molten metal filter shown in FIG. 1.

FIG. 3 is a schematic representation of a micrograph of a cross section of the preferred embodiment of the present invention.

FIG. 4 is a projection view, partially cut-away, of an alternate pore former within the present invention.

DETAILED DESCRIPTION

According to the present invention, a sintered or bonded porous ceramic body containing interconnected cells is formed. The cells are interconnected generally in an open cellular-like mode such that circuitous or tortuous pathways are formed through and throughout the ceramic body. The cell sizes may be altered, in either a gradation or abruptly, from one section of the body to another. It is also possible in a given ceramic body, according to the present invention, to form unconnected cells, or no cells at all, in some portions of the body, while other portions contain interconnected cells, thus localizing the pathways through the ceramic body in certain portions or areas thereof, as desired, as will be well understood by those with skill in the field on reading the following description. It is also possible in a given ceramic body, according to the present invention to include a range of sizes and/or shapes of cells such that either the size range is uniformly repeated throughout the ceramic body from one section thereof to another or the cell sizes are randomly arranged and positioned throughout the body.

The ceramic body of the present invention can be formed and shaped, generally, by any conventional method that is applicable to the shaping and forming of sintered ceramic bodies. FIG. 1 schematically shows a preferred form and shape of the ceramic body 11 which is particularly useful in filtering molten metal. FIG. 1 shows a flat plate form of ceramic body 11 in the general shape of a square, however, the shape of this preferred embodiment could be rectangular, round, hexagonal, irregular, etc., depending upon the shape of the holding mechanism (not shown) into which the ceramic body is to be inserted for use. Of course, the sizing of the preferred embodiment of the ceramic body shown in FIG. 1 would be set to correspond to the size of that holding mechanism and to permit ease of insertion and removal in respect thereto.

The thickness 13 of the ceramic body 11 will depend on the holding mechanism, on the one hand, but also could depend on other factors such as the degree of filtration desired, the location of the ceramic body 11 in the filtering apparatus into which it is inserted, the frequency of use thereof, and/or the location and sizing of the interconnected pores throughout the ceramic body 11. Typically, the ceramic body 11 would, for example, have a thickness 13, as shown in FIG. 2, of about 2" (5.1 cm) and, being generally square, would have, for example, equivalent face dimensions 15, nominally, of about 24" (60.7 cm), 20" (50.8 cm), 17" (43.2 cm), 15" (38.1 cm), 12" (30.5 cm), 9" (22.9 cm) or 7" (17.8 cm). The angle of bevel 17, as shown in FIG. 2, is typically, for example, about 17°, however, this also depends on the corresponding segment of the holding mechanism, associated with the filtering apparatus, with which the bevel angle 17 is to be mated.

In operation, the molten metal is preferably gravity flowed, downwardly, through the ceramic body 11, from entry face 21 to exit face 23, as shown in FIG. 2. Of course, it will be well understood by those with skill in the field that gravity flow is not the only means of flowing molten metal, as pumps for such are available in the market. Bevel 19 functions to both serve as a stationary seat abutment for ceramic body 11 and to permit ease of installation and removal thereof from the corresponding holding mechanism as will be easily understood by those with skill in the field. The ceramic body 11 may be used in conjunction with a gasket means (not shown) interposed between bevel 19 and the corresponding mating segment of the holding mechanism, as is known to those with skill in the field.

Bondable or sinterable ceramic powders, or mixtures of ceramic powders which are bondable or sinterable, are used as the starting raw material. The specification of the powder depends on the application in which the porous ceramic body is to be used. For filtration of molten aluminum, for example, alumina ( $Al_2O_3$ ) may be used with appropriate sintering aids, or binders, such as, for example, calcium-alumino-borate glass or a phosphate-alumino-borate glass. Also particularly useful as sintering aids for alumina are magnesium oxide and/or calcium oxide. A typical composition of a liquid phase sinterable ceramic powder is 97.0 wt. percent  $Al_2O_3$  with the addition of 3.0 wt. percent of glass powder formed from a combination of  $CaO$ ,  $Al_2O_3$  and  $B_2O_3$  (molar ratio = 1:0.79:1.31). A typical composition of a solid state phase sinterable ceramic powder is 99.8 wt. percent  $Al_2O_3$  with 0.2 wt. percent  $MgO$  added. An example of an alumina ( $Al_2O_3$ ) which can be used is Al6SG which is commercially available from the Aluminum Company of America (Alcoa). Other equivalent sources are also available commercially.

For higher temperature applications, such as the filtering of molten copper, molten ductile or grey iron, or molten steel, it is preferred that a sinterable starting powder which is predominantly high purity, submicron partially stabilized zirconia (PSZ), or a mixture of PSZ and spinel powders which are sinterable, be used. Other materials may also be used in such high temperature applications or for other applications of the porous body. Such materials, for example, are  $SiC$ ,  $TiB_2$ ,  $B_4C$ ,  $Si_3N_4$  and  $SiAlON$ , all of which can be rendered sinterable and/or bondable as is well known by those with skill in the field. Materials which are usable as ceramic powders within the present invention may be either liquid phase or solid phase sinterable.

The polymer binder that is used may be either a thermosetting or a thermoplastic organic binder which can be pyrolyzed at temperatures below the sintering or bonding temperature, as the case may be, of the ceramic material. A preferred characteristic of the polymer binder is that it can be thoroughly and easily mixed with both the ceramic powder and the pore former. Preferably, the polymer binder, which constitutes about 30 to 80 volume percent of the mixture which contains only that polymer binder and the sinterable ceramic powder, will have a viscosity of less than about one million centipoise. This relatively low viscosity especially facilitates the blending of the polymer binder into the ceramic powder and pore former materials with the use of conventional mixers such as, for example, double arm mixers or conical mixers. Low viscosity is especially important where the consolidation process includes the injection molding of a paste of the polymer binder and the ceramic powder into a bed of pore former material. In such an application,

the low viscosity decreases the injection pressure required and also the degree of elevated temperature as is normally associated with the injection molding process. Especially low viscosity is preferred for use of the consolidated material in the pour molding or slip casting methods of forming ceramic bodies. Also, it is possible to extrude the consolidated material into various forms.

5 Appropriate plasticizers can be used with the various types of usable polymer binders. For example, mineral oil can be blended with either high or low density polyethylene resins to lower the viscosity, and dibutyl phthalate can be blended with polyester resins with a similar effect. Such plasticizers, when integrated into the resin, may also act as a lubricant and/or may cause or enhance the beading of the resin in contact with the pore former surface.

10 A variety of different polymer binder materials can be used in the present invention. As suggested above, thermoplastic resins such as low or high density polyethylene are suitable, especially where injection molding techniques are used to form the consolidated body. Thermosetting resins such as epoxy or polyester are suitable where die pressing techniques are used to form the consolidated body. Polyester, in particular, is suitable, in relatively low viscosity form, for use where slip casting or pour molding techniques are used to form the consolidated body. Other types of resins are considered useful, such as, for example, polypropylene, phenolic and polyvinylchloride, provided they meet the above stated functional criteria. One key criterion in respect to the resin used is that it must either be liquid state, to begin with, or it must liquefy during the consolidation process, at least on its surface. Following this the resin must solidify. This phenomenon is referred to herein, variously, as "curing", "cure" and "cured".

20 One of the primary requirements for the resin, used as the polymer binder, is that it provide sufficient strength, after curing, to enable handling of the formed green body and, preferably, machining thereof. Another primary requirement is that the polymer binder maintain the structural integrity of the green body during the removal of the pore former materials and to a sufficiently high temperature to ensure that the green body does not disintegrate before it is calcined, as will be explained hereinafter. As used herein, the term "integrity" or "structural integrity" refers to capability of the structure to maintain and sustain itself in the shape and form in which it is produced, without external support, during succeeding manufacturing steps and thereafter in normal handling.

The pore former can be any suitable material that can be readily removed by liquification or other means from the consolidated body. For example, another means of removal of the pore former may be pyrolyzation or direct sublimation. One approach to removing the pore former material by liquification is by leaching with a solvent. An example of this is the use of calcium chloride,  $\text{CaCl}_2$ , as a pore former, with the leaching done, simply, with water. Another method of removing the pore former by liquification is by heating the green body to melt out the pore former material. An example of a pore former that can be used in this method is wax.

35 A critical element in the present invention is that there be a predominant relative difference in the surface tension of the pore former in respect to the surface tension of the resin. That is to say that the wettability of the resin in respect to the pore former should be such that a "beading" effect occurs. Thus, in this respect, the resin, before curing, will tend to bead on those surfaces of the pore former where there is no physical restraint to such; i.e., where there is some discrete open volume in which the beading can occur. As will be described hereinafter this occurs, within the present invention, in two (2) significant areas: the surfaces of the cells and the surfaces and edges of the interconnecting pores. As used herein, the term "edges" in respect to "interconnecting pores" or "pores" refers to the point or discrete area of transition between the cell walls and the walls of the interconnecting pores. The beading effect may be micro, for example, where the cell walls are smoothed by this beading, or it may be relatively macro, for example, where the pore edges are rounded off thereby. The wettability differential may be effected naturally as, for example, where the pore former material naturally produces a beading of the resin. Or it may be affected by modifying the composition of, and/or the surface of, either the pore former or resin, or both, such as for example, by adding a lubricant to the pore former, or adding plasticizers to the resin, or both. As used herein, the terms "lubricating properties", "lubricant", and "lubricity" are related to the foregoing beading phenomena as well as to the normal properties associated with those terms in respect to friction reduction. As used herein, the term "predominantly" means more so, than not, that which is specified, and the term "substantially" means being largely, but not wholly, that which is specified.

50 In the preferred embodiment of the present invention, the pore former is either combined with a lubricant, or acts, itself, as a lubricant. The lubricant can be soaked into the pore former, coated onto the pore former or mixed with the pore former material when the pore formers are, themselves, formed. For example, calcium chloride granules being porous, may be soaked in a lubricant such as, for example, No. 2 diesel fuel oil. As another example, urea spheres have been made by mixing the urea in No. 2 diesel fuel oil, followed by the formation of small sized spherical particles by conventional methods. Wax, on the other

hand, does not normally need the addition of a lubricant as it tends to act naturally as a lubricant by itself. These phenomena may also be found where other materials, for example, low melting point metals such as zinc or lead, are used as pore formers. It has been found that by using either the lubricating properties of the pore former material, or by the combination of a lubricant with the pore formers or the pore former materials, the wall surfaces of the cells formed are noticeably smoother than those formed without a lubricating means. In addition, the ease of formation of the green body in pressing and injection molding techniques, is greatly enhanced. That is to say that the pressure requirements of consolidation, where pressure is used, are significantly reduced.

In addition, or alternatively, the lubricant, upon heating, may generate enough vapor pressure to puncture a thin film of ceramic and polymer which is interposed between adjacent particles of pore former. This will aid in generation of connected porosity. The properties of the lubricant should preferably be such that they allow lubricated pore former particles to contact each other, without any substantial intervening mixture of ceramic and polymer binder at those points where adjacent particles of pore former are in close proximity to each other. This displacement of the ceramic and resin binder is preferred to enhance connected porosity.

Typically, pore formers in the range of +3.5-4, +4-6, +6-8 and +8-20, as related to Tyler mesh screen sizing, are suitable for forming pores in the size range of about 500 to about 1300 microns. These pore former size ranges are specifically useful in a substantially alumina sintered ceramic body for filtering molten aluminum. A preferred pore former material for forming the foregoing sized pores in a sintered alumina body, where leaching is appropriate to remove the pore former, is calcium chloride. Preferably, the calcium chloride is sieved through a series of Tyler mesh screens, as is well known to those with skill in the field, and used, either monosized or multisized, as desired. A readily available and acceptable form of calcium chloride which may be used as a pore former is calcium chloride ice melting material available at most hardware stores for home use, etc. In all pore formers, a generally spherical shape is more preferred, although other shapes which will permit the formation of smooth walled, interconnected cells, are equally acceptable. The surface of the pore formers as used should exhibit a smoothness of about 125 micro inches or smoother, preferably 32 micro inches or smoother for best results. This can be achieved either by employing pore formers with these surface characteristics or by the surface of the pore formers being smoothed by the adherence of liquid during the consolidation process.

An alternate shape of a pore former is shown in FIG. 4. The pore former of FIG. 4 has been formed from a solid, rectangular shaped block of wax 41. The forming has been done by any appropriate means or method as will be well understood by those well skilled in the field. The forming is in the nature of grooves 43 extending half way through the depth 45 of the block 41 from face 49. Alternate grooves 47, set at a 90° angles (perpendicular) to grooves 43, extend half way through the depth 45 of the block 41 from face 51 which is that face which is opposite to, but on a parallel plane with, face 49. At the point where the depth extension of grooves 43 and alternate grooves 47 meet, windows 53 are formed. As will be well understood by those with skill in the art, the pore formers of FIG. 4 may be stacked upon each other with alternate grooves 47 being positioned perpendicular to or otherwise at angle to grooves 43 of the block 41 beneath. Also windows 53 of each block 41 may be aligned or non-aligned with the windows 53 of the block 41 beneath. The pore former shown in FIG. 4 is particularly advantageous where a filter medium of zirconia is to be formed for the filtration of molten ferrous metals such as, for example, steel.

As mentioned above, the mixing of the ceramic powders and resin components of the system can be done in any conventional manner. It is, however, important that those components be uniformly mixed and equally and evenly dispersed among each other to ensure that the sinterable ceramic powder is evenly dispersed in the polymer binders. When the mixture of these components is subsequently blended and mixed with the pore formers, it is important to ensure that there are no voids therein and that all interstitial space between the pore formers is substantially filled to ensure structural strength and integrity.

In a situation where localization, gradation or disruption in the uniform dispersion of the cells (as previously filled by the pore formers) is desired in the sintered porous body, the pore formers are added selectively to the mixture. For example, a uniform mixture of sinterable ceramic powder and polymer binder are formed and divided into two (2) portions. Into the first portion, a given size of pore formers are uniformly mixed and the mixture is placed into the die of a press, partially filling the die cavity, and leveled. The second portion of the sinterable ceramic powder/polymer binder mixture is mixed with a different sized pore former and is placed into the same die cavity, on top of the first mixture, thus filling the die cavity to the appropriate level for pressing. The sintered porous ceramic body resulting from this method will have localized cell sizing, with two distinctly different cell sizes in two different sections of that body, respec-

tively. This method can be varied to add additional cell sizes to localized sections of a given body, or to decrease or increase the number of cells for a given volume at localized sections of that body. Also, it is possible to eliminate the cells in discrete portions or sections of that body, as will be well understood by those with skill in the field.

5 A mixture of Alcoa A16SG alumina was blended with Silmar S-585 polyester resin, as manufactured and marketed by The Standard Oil Company and 0.5 wt. percent of methyl ethyl ketone (MEK) peroxide, manufactured by Penwalt/Ludicol, as a curing catalyst for the polyester resin. The alumina constituted about 40 vol. percent of the mixture of alumina and polyester resin. Added to the mixture of these first two components was calcium chloride of a + 4 - 6 Tyler mesh sizing. The calcium chloride comprised about 70  
10 vol. percent of the final mixture of the three components. The mixture was thoroughly blended in a standard double arm mixer and several portions were subsequently loaded into an 2 1/4" (5.7 cm) round die, with a 1" (2.5 cm) working depth, and pressed. It was found that consolidation die pressures in a range of 2400 to 2500 psi were adequate to ensure that the calcium chloride granules were sufficiently in contact with each other to produce the formation of substantially interconnected pores in the sintered porous ceramic body produced. As used herein, the term "cells" refers to the discrete open volumes, or voids, within the porous ceramic body formerly occupied by the pore formers and the term "pores" refers to the interconnecting passageways between the cells. This polyester resin was selected for use with calcium chloride because it was noted that it appeared to have a relatively low amount of wettability in respect to calcium chloride. That is to say that it appeared to bead up more when in contact with the surface of a calcium chloride pore  
15 former in comparison to other polyester resins which were considered.

It has been found that the pressure required to consolidate the green bodies being formed can be significantly reduced if a lubricant is added to the pore former or if the pore former is self-lubricating, as is the situation where wax pore formers are used. This reduction in pressure is significant, being about 1/3 or less of that otherwise required. In the press and die compaction forming technique, the typical 2400 psi to  
20 2500 psi pressure, normally required to compact the mixtures to the point where the sintered porous body contained cells that were substantially interconnecting, was capable of being reduced to a typical range of about 700 psi to 800 psi. This was done by impregnating the porous calcium chloride with No. 2 diesel fuel oil. Another benefit accrued, by using the lubricant, is that the walls of the cells of the sintered porous body were visibly and notably smoothened. Further, it was found that the edges of the pores were rounded off and smoothened. And further, it was found that the walls of the pores were notably smoothened. The significance of this is that there is less resistance to the flow of the molten metal through the porous body. The term "consolidation" as used above in this paragraph refers to the application of mechanical pressure to the blended mixture of ceramic powders, resins and pore formers, causing substantially all of the pore  
25 formers to be in sufficiently close spatial relationship to each other such that pores develop between each cell and at least two (2) adjacent cells, following liquification of the pore formers and sintering. Otherwise, "consolidation" refers to any method by which the mixture of ceramic powder and resin is interposed between and among the pore formers such that pores develop between each cell and at least two (2) adjacent cells. For example, in using pore formers as shown in FIG. 4, the viscosity of the resin/ceramic mixture is reduced to the point where it can be pour molded by selection of low viscosity resins and/or by  
30 the addition of appropriate plasticizers as will be understood by those with skill in the field.

Once the green body has been formed by way of consolidation, as described above, and the resin has cured, the pore former must be removed. When a leachable pore former is used, after the resin in the green body has cured, the green body is immersed into a solvent to dissolve the pore former. For example, when calcium chloride is used as the pore former, the resin cured green body is immersed in water for a  
35 sufficient time to dissolve the calcium chloride and leave a porous green body. The porous green body is then preferably thoroughly rinsed with water, to remove traces of calcium chloride, and dried. Of course, for the pore formers to dissolve and leave, the pores must have been formed.

When wax or another meltable pore former is used, that pore former must be melted out by subjecting the green body to a temperature above the melting point of the pore former material. This can be done as a  
40 separate step or the pore former can be melted out as the green body is gradually raised to calcining and then sintering temperature as will be explained hereinafter. When using wax as the pore former, some of the pores are formed by direct contact of the pore formers with each other after consolidation, while other pores may be formed by the gas pressure created by the volatilization of the hot wax as the temperature of the green body is increased toward full sintering temperature.



The shaped, resin-cured green bodies, consisting of the polymer binder and the sinterable ceramic powder, are then fired in a furnace to both burn off the polymer binder and to sinter or bond the ceramic powder into a porous ceramic body. As the polymer binder burns off, the ceramic powder particles move towards each other, densifying as the particles sinter to each other. The overall dimensions of the ceramic body decrease as this occurs, producing a dimensional "shrink" and the sintering produces high structural strength and integrity.

Typically, the green body is placed in a furnace which can be controlled to vary the rate of increase in temperature per unit of time. When the sinterable ceramic powder is substantially alumina, the temperature of the furnace, for example, beginning at about 150°C, is raised to about 600°C at a rate of about 10°C/hour to 30°C/hour. This step effectively calcines the green body. No special atmosphere is needed as sintering of alumina will occur in air. At the point where the green body has reached 600°C, any meltable pore former should have either melted out or vaporized. And most, if not all, of the polymer binder has burned off at 600°C, and the densification and shrinkage commences thereafter as the temperature is increased to full sintering temperature.

At the point where the substantially alumina green body, substantially minus the pore former, has reached, for example, about 600°C, the rate of temperature elevation may be, for example, increased to a range of about 200°C/hour to 400°C/hour, and the green body is brought up to full sintering temperature and held there for a sufficient time for complete sintering to occur. For solid state phase sintering of Alcoa Al6SG alumina, with a 0.2 wt. percent MgO added, a sintering temperature of 1550°C for two hours is generally considered sufficient for substantially complete sintering. The sintering produces a mechanically strong, self-sustaining body with cells substantially interconnected by pores, with relatively high impact resistance and relatively substantially reduced friability in comparison to known ceramic filters.

After a substantially alumina ceramic body has been held at the sintering temperature for a sufficient time to substantially sinter the ceramic material, the body must be cooled. It is, of course, economically expedient to cool the body as quickly as possible. On the other hand, care must be taken to cool it slowly enough so that thermal shock and cracking are avoided. Once cooled, the porous ceramic body is ready for use.

In mixing the components together for use in injection molding techniques, it is preferable to leave out the pore formers from the initial component mix, thus only mixing together the sinterable ceramic powder and the polymer binder. The pore formers, however, are loaded into the cavity of the injection molding apparatus and the sinterable ceramic powder/polymer binder mixture is then injected into that cavity to fill the interstitial space between the pore former particles, thus simultaneously admixing the resin/ceramic powder mixture with the pore formers and consolidating those components.

Alternatively, in respect to injection molding, the sinterable ceramic powders and the polymer binders can be mixed together in such volumes that the sinterable ceramic powders constitute between 20 and 70 vol. percent of that mixture and, preferably, between 40 and 70 vol. percent thereof. Using thermoplastic resins as polymer binders, these mixtures can be granulated using standard procedures for compounding polymers with inert fillers. The granules are then injected into the die of an injection molding apparatus, provided, that the die has first been loaded with pore formers and the pore formers pressed into a compact bed just before the injection of the granules. It has been found that this method ensures that the pore formers are all in contact with each other and that all of the interstitial space between the pore formers are filled with the sinterable ceramic powder/polymer binder mixture. The pieces are otherwise molded according to standard injection molding practice. The sintered porous ceramic bodies thus produced have been found to include substantially interconnected cells.

It has also been found that granulated mixtures of sinterable ceramic powder and thermoplastic resin can be blended with pore formers and pressed together in a standard compression molding apparatus wherein the die is heated to a temperature selected according to the viscosity/temperature relationship of the particular resin being used.

When thermosetting resins are used, the pieces can be cured in the die by heating the die following compaction, or the pressed pieces can be removed directly from the die following pressing and cured in a separate step. In the case of the use of thermoplastic resins, the pieces are allowed to cool and cure in the die, so as to facilitate their removal from that die.

## EXAMPLES

Example 1. A set of four (4) porous sintered ceramic bodies were fabricated, two of which contained 97 wt. percent Alcoa Al6SG alumina and 3 wt. percent calcium-alumino-borate glass powder and two of which contained 99.8 wt. percent Alcoa Al6SG alumina and 0.2 wt. percent magnesium oxide (MgO). The details of fabrication are as follows:

Sample Code:	Ceramic: Vol. per- cent of Ceramic plus resin mixture	CaCl <sub>2</sub> : pore for- mer Vol. Ceramic, resin and pore for- mer	Density: g/cc after leaching	Density: g/cc after sintering
A6040	40.5	66	0.67	0.90
A7040	40.5	75	0.50	0.66
B6040	39.7	66	0.70	0.91
B7040	39.7	75	0.48	0.62

A6040 and A7040: 99.8 wt. percent  $\text{Al}_2\text{O}_3$ , 0.2 wt. percent MgO

B6040 and B7040: 97.0 wt. percent  $\text{Al}_2\text{O}_3$ , 3.0 wt. percent glass ( $\text{CaO}/\text{Al}_2\text{O}_3/\text{B}_2\text{O}_3 = 1/0.79/1.31$  Molar Ratio) manufactured by Ferro Corporation and designated as XF 41

Pore Former: Average diameter, 3.2 mm.,  $\text{CaCl}_2$ , J. T. Baker Desiccant grade

Resin: Epoxy (Epokwick) + Hardener (EpoKwick) both marketed by Beuhler

Compaction Pressure: ~ 2500 psi @ room temperature

Sample Diameter: 2.25"

Cure: Room Temperature ~ 2 hours

Leaching: Water @  $90^\circ\text{C} \pm 10^\circ\text{C}$  for ~ one hour

Drying: Optional

Firing:  $150^\circ\text{C} - 800^\circ\text{C}$  @ ~  $20^\circ\text{C}/\text{hour}$   
 $800^\circ\text{C} - 1550^\circ\text{C}$  @ ~  $125^\circ\text{C}/\text{hour}$   
 $1550^\circ\text{C}$  hold for 2 hours  
 $1550^\circ\text{C} - 20^\circ\text{C}$  furnace cool, ~ 6 hours

Drying was done between leaching and forming only to enable green body density measurements; drying is not required to make the sintered body. The particular epoxy resin/hardener system was selected as it appeared to bead to a relative greater degree, in contact with  $\text{CaCl}_2$  pore formers, than some other epoxy/hardener systems that were considered.

After sintering, the sample pieces were measured to determine the effects of sintering on the density and sizing of the green bodies as measured before firing. The results are as follows:

Sample Code:	Density: g/cc after sintering	Percent change in density from green body to sintered body	Diameter Change: percent	Thickness Change: percent
A6040	0.896	+35.3	-19.9	-17.7
A7040	0.662	+32.9	-19.7	-18.1
B6040	0.910	+30.2	-18.7	-19.1
B7040	0.622	+28.6	-18.9	-19.4

The sintered porous ceramic sample pieces were then examined to determine the average pore sizes. Those same sample pieces were then tested to determine, firstly, the starting head of molten aluminum that was maintained thereby. That is to say, the depth of molten aluminum above each filter body sample piece was measured. Secondly, the percentage volume of the filter body, which was filled by molten aluminum, was determined. The third item which was determined, but prior in time to the molten aluminum tests, was the volume of air per unit of time that would flow through the filter body at a standard air pressure of 2000 dynes/cm<sup>2</sup>, Re > 20, against the entry face of the filter body sample piece. The results of these tests are as follows:

Sample Code:	Average Sintered Pore Diameter	Initial Metal Head Depth	Vol. % Filled by Al	Air Flow Rate cm <sup>3</sup> /sec at 2000 dynes/ cm <sup>2</sup>
A6040	1.140mm	11.8cm	45.6	99
A7040	1.245mm	11.0cm	40.9	123
B6040	0.840mm	15.6cm	48.6	74
B7040	0.560mm	24.3cm	39.3	66

EXAMPLE 2. A paste was prepared by mixing epoxy resin, a hardener and  $Al_2O_3 + 0.2\%$  MgO in a ratio such that the ceramic was 40 volume percent of the paste. The epoxy resin and hardener were those that were stated, above, for Example 1, and the alumina ( $Al_2O_3$ ) was the Alcoa Al6SG material, also as stated above for Example 1. This paste was injected into the cavity of an injection molding apparatus after first having packed the 3" diameter die of that injection molding apparatus with +4-6  $CaCl_2$  granules. The injection pressure used was 1600 psi. After substantial infiltration of the paste into the interstitial space between the packed  $CaCl_2$  granules, the consolidated piece was removed from the die and the resin cured at room temperature. The piece was then leached in water and dried. The piece was then fired according to the firing schedule defined, above, for Example 1, and cooled. The shrinkage, comparing the green body with the sintered body, was 21% for the diameter and 27% for the thickness. The final density of the sintered body was 1.15 g/cc.

EXAMPLE 3. A set of two (2) porous sintered ceramic bodies were fabricated which contained 99.8 wt. percent Alcoa Al6SG alumina and 0.2 wt. percent MgO. The details of the fabrications are as follows:

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Sample Code:	Ceramic: Vol. per- cent of Ceramic plus resin mixture	CaCl <sub>2</sub> : pore for- mer Vol. %, Ceramic, resin and pore for- mer mixture	Density: g/cc after leaching	Density: g/cc after firing
A7060- 46-05	40	70	0.51	0.73
A7060- 46-06	40	70	0.55	0.80

A7060-46-05

A7060-46-06: 99.8 wt. percent Al<sub>2</sub>O<sub>3</sub>, 0.2 wt. percent MgO

Pore Former: -4 + 6 mesh (Tyler) size, CaCl<sub>2</sub>

Resin: Standard Oil Co. Silmar S-585 polyester resin with  
0.5 wt. % of Pennwalt Ludicol DDM-9 MEK peroxide  
added as a curing catalyst

Compaction A7060-46-05 = 750 psi @ room temperature.

Pressure: A7060-46-06 = 1250 psi @ room temperature

Sample A7060-46-05 = 2.25" dia x 0.98" thick

Diameter: A7060-46-06 = 2.25" dia x 0.90" thick

Cure: ~ 55°C for 1/2 hour, die cured

Leaching: Flowing water in a tank @ 70-75°C for ~ 1 hour  
until weight loss had reached a constant value

Drying: Optional

Firing: 20°C - 600°C, 29 hrs. @ ~ 20°C/hour increase  
600°C - 1550°C, 4.75 hrs. @ ~ 200°C/hour increase  
1550°C hold for ~ 2 hrs.  
1550°C - 20°C, ~ 4 hrs., furnace cool

The resin (plus catalyst) was first mixed to uniformity. To this were added the  $Al_2O_3$  and  $MgO$  powders. This combination was then mixed sufficiently to ensure homogeneity and complete wetting of the powders. Then the pore former  $CaCl_2$  was added and a third mixing step was employed, again to ensure homogeneity and complete admixing of the  $CaCl_2$  with the resin/ceramic powder mixture. The mixing times for each of the three mixing steps was in the range of 1-3 minutes.

The mixture was then removed from the mixer and was loaded into a standard steel die, using a combination of vegetable oil and wax paper as a mold release. The steel dies were then mounted in a Carver hydraulic press and compacted with a dwell time of 10-15 seconds. Then the samples were cured in the die at ~ 55°C for one-half hour then removed. Following the curing, the samples were placed in a water tank, through which water was flowing, and leached until the weight loss had reached a constant value, to remove the pore former. Then the samples were dried for a sufficient time to evaporate substantially all of the water from the pores created by leaching. Finally, the dried body was fired using the Schedule A cycle, described above in this Example.

After sintering, the sample pieces were measured to determine the affects of sintering on the density and sizing of the green bodies as measured before firing. The results are as follows:

Sample Code:	Density: g/cc after sintering	Percent change in density from green body to sintered body	Diameter Change: Percent	Thickness Change: Percent
A7060-46-05	0.73	+45.0	-22.0	-21.9
A7060-46-06	0.80	+45.5	-20.5	-22.8

The sintered porous sample pieces were then examined to determine the average pore sizes. Those same sample pieces were then tested to determine, firstly, the starting head of molten aluminum that was maintained thereby. Secondly, the percentage volume of the filter body, which was filled by molten aluminum as it flowed through that filter body, was determined. Finally, the aluminum permeability, in lbs./sq. ft./min. was determined by general application of the following formula:

$$\frac{\text{lbs. of Al}}{\text{Sq. ft./min}} \times \frac{\text{in (thickness)}}{\text{pressure head (in Al)}}$$

The results are as follows:

Sample Code:	Average Sintered Pore Diameter	Initial Metal Head Depth	Vol. % Filled by Al (Porosity)	Aluminum Permeability lb/sq. ft. min.
A7060- 46-05	0.031"	6.9"	69	60
A7060- 46-06	0.048"	4.4"	51	190

It will be noted that the compaction pressures used in Example 3 were significantly reduced from those used in Example 1, this being attributed to the relatively greater lubricating properties exhibited by the resin used in Example 3 in comparison to that used in Example 1.

EXAMPLE 4. A set of two (2) porous sintered ceramic bodies were fabricated using a fabrication technique similar to those of Example 3, the only significant difference being that the  $\text{CaCl}_2$  pore former was soaked in #2 diesel fuel oil prior to incorporation into the resin/ceramic powder/pore former mixture. The details of the fabrication of these two (2) sample batches are as follows:



Sample Code:	Ceramic: Vol. per- cent of Ceramic plus resin mixture	CaCl <sub>2</sub> : pore for- mer Vol. Ceramic, resin and pore for- mer mixture	Density: g/cc after leaching	Density: g/cc after firing
OA7060- 46-01	40	70	0.54	0.69
OA7060- 46-08	40	70	0.63	0.75

The compaction pressure used to form green body OA7060-46-01 was 750 psi and to form green body OA7060-46-08 was 1250 psi. The sample size of OA7060-46-01 was 2.25" diameter X 0.90" thick, and the sample size of OA7060-46-08 was 2.25" diameter X 0.86" thick. The firing schedule used for the samples of both Example 4 and Example 5, following, is as follows:

60°C - 600°C, 36 hours, ~ 15°C per hour  
 600°C - 1550°C, 4.75 hours, ~ 200°C per hour  
 1550°C hold for 2 hours  
 1550°C - 20°C, ~ 4 hours

After sintering, the sample pieces were measured to determine the affects of sintering on the density and sizing of the green bodies as measured before firing. The results are as follows:

Sample Code:	Density: g/cc after firing	Percent change in density from green body to sintered body	Diameter Change: Percent	Thickness Change: Percent
OA7060- 46-01	0.69	+26.8	-20.7	-21.3
OA7060- 46-08	0.75	+19.7	-20.9	-20.5

The sintered porous sample pieces were then examined to determine the average pore sizes. Those same sample pieces were then tested to determine, firstly, the starting head of molten aluminum that was maintained thereby. Secondly, the percentage volume of the filter body, which was filled by molten aluminum as it flowed through that filter body, was determined. Finally, the aluminum permeability, in lbs/sq. ft./min. was determined. The results are as follows:

Sample Code:	Average Sintered Pore Diameter	Initial Metal Head Depth	Vol. % Filled by Al (Porosity)	Aluminum Permeability lb/sq. ft. min.
OA7060- 46-01	0.053"	4.0"	63	483
OA7060- 46-08	0.064"	3.3"	61	542

It should be restated and emphasized that the compaction pressure necessary in Example 4 was significantly lower than that which was necessary in Examples 1 and 3, the significant change being that a lubricant was used with the pore formers in Example 4 while none was used in Examples 1 and 3. In Example 3 where the compaction pressures were parallel, sample-for-sample, with those of Example 4, the aluminum permeability was greatly increased. The only significant change between Example 3 and Example

4 was the addition of a lubricant to the pore former of Example 4 despite the fact that an epoxy resin, with relatively lower lubricating characteristics, was used in Example 4. Also, a comparison of the cells and the pores of the samples of Examples 1 through 3 with the samples of Example 4 showed a significant increase in the smoothness of the cell walls and the pore walls, and a much greater smoothing or rounding of the edges where the cell walls met the pores for the samples of example 4.

**EXAMPLE 5.** A set of two (2) porous sintered ceramic bodies were fabricated which were identical in fabrication technique to those of Example 3 except that  $\text{CaCl}_2$  was not used as a pore former material. Rather, wax pore formers were prepared from Kindt-Collins #KC210 wax, having a melting point of about  $90^\circ\text{C}$ , sufficiently above the  $70^\circ\text{C}$ - $75^\circ\text{C}$  curing temperature of the resin so as to remain solid during the curing step of the process. As with the  $\text{CaCl}_2$  pore formers used in previous examples, the wax pore formers were sized at  $-4 + 6$  mesh (Tyler) size, by standard sieving techniques employing Tyler mesh screening. And, as a consequence of using wax, instead of  $\text{CaCl}_2$ , as a pore former, the leaching step was replaced with a melting step, following the resin curing step. The resin-cured green body was heated to about  $110^\circ\text{C}$ , somewhat above the melting point of the wax, to liquefy the wax, thus causing it to flow out of the green body, creating the pores. The resin cured green body was maintained at this temperature for a time sufficient to permit the weight loss to reach a constant value. In all other respects, the fabrication techniques applied were equivalent to those of Example 1. The details of the fabrication of these two (2) samples are as follows:

Sample Code:	Ceramic: Vol. per-cent of Ceramic plus resin mixture	Wax: Vol. Percent Ceramic, resin, and pore former mixture	Density: g/cc after wax melt-out	Density: g/cc after firing
WA7060-46-05	40	70	0.60	0.75
WA7060-46-06	40	70	0.61	0.75

The compaction pressure to form green body WA7060-46-05 was 750 psi and to form green body WA7060-46-06 was 1250 psi. The sample size of WA7060-46-05 was 2.25" diameter  $\times$  0.92" thick, and the sample size of WA7060-46-06 was 2.25" diameter  $\times$  0.91" thick.

After sintering, the sample pieces were measured to determine the affects of sintering on the density and sizing of the green bodies in comparison with those measurements before firing. The results are as follows:

Sample Code:	Density: g/cc after firing	Percent change in density from green body to sintered body	Diameter Change: percent	Thickness Change: percent
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WA7060- 46-05	0.75	+23.9	-20.9	-21.8
WA7060- 46-06	0.75	+22.5	-21.2	-21.8

The sintered porous sample pieces were then examined to determine the average pore sizes. Those same sample pieces were then tested to determine, firstly, the starting head of molten aluminum that was maintained thereby. Secondly, the percentage volume of the filter body, which was filled by molten aluminum as it flowed through that filter body, was determined. Finally, the aluminum permeability, in lbs/sq. ft. min. was determined. The results are as follows:

Sample Code:	Average Sintered Pore Diameter	Initial Metal Head Depth	Vol. % Filled by Al (Porosity)	Aluminum Permeability lb./sq. ft. min.
WA7060- 46-05	0.056"	3.75"	71	560
WA7060-	0.062"	3.40"	70	625

A significant portion of the cells in this Example were deformed, more or less in the form of squashed spheres, whereas in previous Examples, the cells were found to be substantially spherical. It is believed that the deformed cells were caused by the relative softness of the wax pore formers and the high malleability of wax under pressure.

As will be noted from comparing Examples 3-5, there is a significant increase in aluminum permeability between Example 3, where a  $\text{CaCl}_2$  pore former was used without a lubricant, and Example 4, where  $\text{CaCl}_2$ , soaked in #2 diesel fuel oil as a lubricant, was used as a pore former. Also there is a significant increase in aluminum permeability between Example 4, where lubricated  $\text{CaCl}_2$  was used as a pore former and

Example 5 where self-lubricating wax was used. Visual inspection of the cell walls, the pore walls and the pore edges of Examples 3-5 revealed that those of Examples 4 and 5 appeared much smoother than those of Example 3, with the pore edges of the samples of Examples 4 and 5 being much more rounded and smoothened than those of Example 3. A comparison of the samples of Example 4 with Example 5 showed that the samples of Example 5 had smoother cell and pore walls and the pore edges of the samples of Example 5 appeared more rounded and smooth. The pore diameters of the samples of Example 3 were significantly smaller than those of Examples 4 and 5 although the same size of pore formers were used in the samples of all three of Examples 3-5. Also there is a significant drop in the initial aluminum metal head depth in comparing the samples of Example 3 with those of Examples 4 and 5. In comparing the volume percentage of the body filled by molten aluminum; i.e., the effective porosity, of the bodies of Examples 3-5, it is noted that sample A7060-46-05 of Example 3 had a greater percentage (68%) of porosity than sample OA7060-46-05 of Example 4 (63%) while the pore diameter of sample A7060-46-05 was less (0.031") than that of OA7060-46-05 (0.053"); yet the starting head of A7060-46-05 was significantly greater than OA7060-46-01 and the aluminum permeability of A7060-46-05 was less than OA7060-46-01. There was no significant difference in the fired or sintered density of the samples of Examples 3-5.

**EXAMPLE 6.** (Sample Code U7060-46-1) A porous sintered ceramic body was made using urea as a pore former. For this sample, Standard Oil Silmar DL-459 polyester resin was used (mixed with Lupercio AMS Hardener as a curing catalyst) along with a ceramic powder mixture. The ceramic powder mixture was 90 wt. percent Alcoa Al6SG Al<sub>2</sub>O<sub>3</sub> plus 10 wt. percent Ferro XF-41 CAB glass frit (CaO/Al<sub>2</sub>O<sub>3</sub>/B<sub>2</sub>O<sub>3</sub> = 1/0.79/1.31 molar ratio). 60 vol. percent of resin was mixed with 40 vol. percent of ceramic powder. This mixture was then mixed with + 6 mesh (Tyler) size urea pore former such that 60 vol. percent of the total mixture was pore former. The total mixture was pressed at 750 psi and cured. The urea pore former was melted out at 150°C. The resultant green body was fired according to the following schedule:

80°C - 800°C @ ~ 15°C/hour, 36 hours  
600°C - 1200°C @ ~ 200°C/hour, 3 hours  
1200°C hold for 2 hours  
1200°C - 700°C @ ~ 1000°C/hour, 0.5 hours  
700°C - 20°C @ ~ 340°C/hour, 2 hours

The fired body of Example 6 was not tested with molten metal (aluminum), however, it appeared visually to be quite similar to the samples of Example 3, above, which were successfully tested with molten aluminum.

**EXAMPLE 7.** A set of five (5) porous ceramic bodies were fabricated using #2 diesel fuel oil as a pore former lubricant, and using a vacuum to evacuate the CaCl<sub>2</sub> pore former to enhance soaking by the lubricant. In forming these porous bodies, the compaction pressure was varied on some to determine if such would have an effect on green body density. The details of fabrication are as follows:

Sample Code:	Die Pressure psi	Density After Pressing	Density g/cc after leaching	Density g/cc after firing	Total Porosity Vol. %
8-7A	750	1.74g/cc	0.71	0.58	85.0
8-7B	750	1.63g/cc	0.59	0.55	86.0
8-7C	750	1.75g/cc	0.66	0.61	84.5
8-7D	1500	1.72g/cc	0.65	0.55	86.0
8-7E	3000	1.81g/cc	0.69	0.63	84.0

These samples were formed with lubricated pore formers using the following mixture to make a single batch from which each sample was formed:

1300g  $\text{CaCl}_2$ , -4 + 6 mesh  
 480.6g Alcoa Al<sub>2</sub>SG alumina ( $\text{Al}_2\text{O}_3$ )  
 24.82 Ferro XF-41 CAB (calcium aluminoborate) glass frit  
 ( $\text{CaO}/\text{Al}_2\text{O}_3/\text{B}_2\text{O}_3 = 1/0.79/1.31$  Molar Ratio)

227.5g Standard Oil S-585 Silmar Polyester resin  
 1.14 g. MEK peroxide, designated DDM-9, manufactured by Ludicol Division of Pennwalt

A standard double arm mixer was used to thoroughly blend the mix. An 8" diameter die was used to form sample no. 8-7A while a 2.25" diameter die was used to form all of the other samples. All of the samples were leached in water for about 3 hours each. The firing schedule for these samples was as follows:

60°C - 600°C @ ~ 15°C/hour  
 600°C - 1200°C @ ~ 200°C/hour  
 1200°C - hold for 2 hours  
 1200°C - 700°C @ ~ 1000°C/hour  
 700°C - 20°C @ ~ 340°C/hour

Samples 8-7C, 8-7D and 8-7E, above, were formed to determine the effect on density by increasing the die pressure; these samples were not further tested. It is noted that there was no significant increase in density after pressing when the die pressure was increased, indicating that full pressed density was achieved at about 750 psi.

One of the lubricated pore former samples, sample number 8-7B, was then compared with sample numbers A7060-46-05 and A7060-46-06, from Example 3, which were made with unlubricated pore formers. The results of these comparisons are as follows:

Sample Code:	Die Pressure psi	Initial Metal Head Depth	Pore Diameter (Range ± 5%)	Vol. % Filled by Al (Porosity) min	Aluminum Perme- ability lb./sq. ft.
8-7B	750	8.39cm	1.65mm	67	1000
A7060-46-05	750	17.53cm	1.22mm	69	60
A7060-46-06	1250	11.18cm	1.22mm	51	190

Note that the pore diameter is significantly increased by the use of lubricated pore formers, although the pore former size ranges and distribution of all three samples were equivalent. The resistance to the flow of molten metal, as measured by the depth of metal head initially maintained by the filter body, is significantly decreased for sample 8-7B. Even when the compaction pressure is doubled (compare A7060-46-05 and A7060-46-06), the initial metal head depth is still significantly greater when compared to the sample made at lower compaction pressure but with a lubricated pore former (compare A7060-46-06 and 8-7B).

Example 8. A set of four (4) porous ceramic bodies were fabricated using zirconium oxide (zirconia,  $ZrO_2$ ) powder as a base ceramic powder. Yttrium oxide ( $Y_2O_3$ ) powder was added to the  $ZrO_2$  powder to produce a ceramic powder mix and the end product ceramic bodies were characterized as partially stabilized zirconia (PSZ). The ceramic powder mix including 3 mole % of  $Y_2O_3$  mixed with the balance of  $ZrO_2$  was obtained commercially from Toyo Soda, designated TZ-3Y. The resin used was Standard Oil S-585 Silmar, polyester resin, using Ludicol MEK peroxide as a curing catalyst. The pore former was +4-6 mesh (Tyler) size  $CaCl_2$  soaked in #2 diesel fuel oil. The details of fabrication are as follows:

Sample Code:	Ceramic: Vol. per-cent of Ceramic plus resin mixture	$CaCl_2$ : pore for-mer Vol. %, Ceramic, resin and pore for-mer mixture	Density: g/cc after leaching	Density: g/cc after firing
Z-3A	30.7	64.2	0.80	1.42
Z-3B	30.7	64.2	0.80	1.46
Z-5A	33.1	65.9	0.85	1.63
Z-5B	33.1	65.9	0.75	1.39

All of the foregoing  $ZrO_2$  samples were compaction pressed at 1500 psi. All of the green bodies produced were fired according to the following schedule:

60°C - 600°C, 36 hrs. @ ~ 15°C/hour increase  
 600°C - 1550°C, 4.75 hrs. @ ~ 200°C/hour increase  
 1550°C hold for ~ 2 hrs.  
 1550°C - 20°C, ~ 4 hrs., furnace cool

The firing was done in an air atmosphere. All of the mixing leaching and curing done for the samples of this Example were done according to the specifications set forth for Example 4, above. The size of the samples produced for this Example were all 2.25" diameter with thickness in the range of 0.47"-0.71".

After sintering, the sample pieces were measured to determine the affects of sintering on the density and sizing of the green bodies as measured before firing. The results are as follows:

Sample Code:	Density: g/cc after sintering	Percent change in density from green body to sintered body	Diameter Change: percent	Thickness Change: percent
Z-3A	1.42	+77.9	-27.1	-24.2
Z-3B	1.46	+82.3	-27.4	-27.0
Z-5A	1.63	+92.9	-27.2	-27.3
Z-5B	1.39	+86.6	-26.5	-26.4

The fired bodies of Example 8 were not tested with molten metal (aluminum), however, they appeared visually to be quite similar to the samples of Example 4, above, which were successfully tested with molten aluminum. In addition, it is believed that because of the ability of PSZ to withstand significantly higher temperatures of molten metals beyond that of molten aluminum, the PSZ samples of Example 8 may be used to filter molten copper, brass, bronze, and steels (both mild steels and stainless steels).

One factor that appears to remain consistent in regard to all of the samples tested and analyzed in the foregoing examples is that the volume of the cells, created in the porous ceramic bodies, is directly proportional to the volume of the pore formers initially included in the consolidated mixtures. In all cases, the volume is reduced but only by the shrink factor associated with sintering which also affects all other dimensions of the porous ceramic body in comparison to the corresponding green body before sintering. Also, another factor that appears to remain consistent throughout those samples is that the shape of the pores is predominantly round. That is to say that more than 50% of the pores have a rounded shape generally approaching the shape of a hollow cylindrical section, as distinguished from being polyhedral in shape while a substantial (more than 25%) additional portion of those pores which are not rounded in shape, are generally oval or elliptical in cross section, as distinguished from having distinct polygon cross sections. Relatively few of the pores, in comparison, are generally irregular in shape such that distinct polyhedrons are formed thereby.

Several samples, similar to the foregoing samples, but which were not tested with molten metal, were tested to determine strength. Two specific types of standard tests were performed to determine, respectively, Modulus of Rupture (MOR) and Modulus of Bending (MOB). At room temperature, the samples tested all exhibited an MOR within the range of 200-260 psi, and at elevated temperature (1200°F) an MOR within the range of 110-160 psi. At room temperature, the samples tested all exhibited an MOB within the range of  $5.5 \times 10^4$  -  $7.2 \times 10^4$  psi, and at elevated temperature (1200°F) an MOB within the range of  $4.2 \times 10^4$  -  $4.7 \times 10^4$  psi. All of the samples so tested for strength were fabricated according to the method stated in Example 5 above except that the volume percent of the ceramic powder used to form the ceramic plus resin mixture was 45 vol. % and the volume percent of the wax pore formers used to form the overall mixture, including pore formers plus ceramic plus resin was 75 vol. %. Based on the foregoing, the conclusion reached was that the porous ceramic bodies of the present invention had sufficiently high enough structural strength and integrity to qualify them as acceptable molten metal filters.

Referring to FIG. 3, there is shown a schematic representation of a micrograph of a cross section of a porous ceramic body 25 made using  $Al_2O_3$  as the ceramic material and wax as the pore former. The degree of assimilated magnification relative to actual size is about 6X. Pores 27, seen as voids in their top view alignment shown in FIG. 3, are predominantly round in shape while some tend toward being elliptical or oval



shaped pores 29. In this embodiment, generally spherically shaped wax pore formers are used, thus many of the cells 31 are spherical in shape while others take the form of squashed spherical cells 33 as explained previously. Generally horizontally arranged pores 35 interconnect the cells 31, 33. Substantially all of the cells are interconnected to at least two (2) other cells. Also, the top view pores 27, 29 interconnect cells beneath (not shown) to those that are shown in FIG. 3. Finally, separating the cells is a dense, substantially solid (void-free) sintered ceramic matrix 37 which, except for the interconnection of the pores 27, 29, 35, substantially completely fills the interstices between the cells 31, 33. This ceramic matrix 37 is depicted by the hatched sections of FIG. 3.

In FIG. 3, substantially all of the pore edges 39 which form the discrete areas where there is a transition from a cell wall 40 to a pore wall 38 are rounded off or smoothed such that there are no sharp corners or rough edges to cause the smooth flow of molten metal to be hindered or restricted. Surprisingly, however, the exogenous intermetallic substances, and other impurities including slag and dross, are trapped within the cells 31 and 33 during the flow of molten metal through the ceramic body 25 with relatively insubstantial amounts passing therethrough. Those exogenous intermetallic substances were found to be trapped substantially uniformly within the cells 31 and 33. Thus, those non-metallic substances were found to be substantially uniformly distributed through the cross section or thickness 13 of the ceramic body 11 as viewed in FIG. 2 extending from entry face 21 to exit face 23 whereas, in most prior art filter media, the exogenous intermetallic substances, after molten metal flow, are found to be collected at or close to the entry face thereof.

According to the Patent Statutes, the best mode and preferred embodiment of the present invention have been described. However, the scope of the present invention is not limited thereto, but rather, is defined by the appended claims.

## Claims

1. A porous ceramic body of high strength and integrity comprising:
  - a.) a substantially continuous and solid ceramic matrix;
  - b.) a plurality of cells distributed and interspersed throughout said ceramic matrix, said ceramic matrix which substantially fills the interstices between and among said cells;
  - c.) a plurality of pores interconnecting said cells such that predominantly each of said cells is interconnected by said pores to at least two (2) other of such cells, the walls of said cells and said pores which are predominantly smooth, the edges of said pores, which form the discrete areas of transition between said cell walls and said pore walls, being rounded and smoothed, said average cross sectional area of said pores which are predominantly smaller in size than the average cross sectional area of said cells.
2. The invention of claim 1 wherein said ceramic matrix is selected from the group comprising  $Al_2O_3$ , SiC,  $TiB_2$ ,  $B_4C$ ,  $Si_3N_4$ , SiAlON and PSZ.
3. The invention of claim 1 wherein the said porous ceramic body is a filter medium.
4. A green body, from which a porous ceramic body of high strength and integrity can be formed, comprising a consolidation of a predominantly homogenous mixture of a curable resin with a sinterable ceramic powder which is intermixed with a plurality of pore formers, the materials of said resin and said pore formers being selected such that the wettability of said resin in respect to said pore formers is such that said resin tends to bead when in contact with the surfaces of said pore formers, before said resin is cured, said resin in said consolidation which is cured and said pore formers in said consolidation which are removable after said resin in said green body has been cured.
5. The invention of claim 4 wherein said curable resin is selected from the group comprising polyester, epoxy, polyethylene, polypropylene, phenolic and polyvinylchloride.
6. The invention of claim 4 wherein said ceramic powder is selected from the group comprising  $Al_2O_3$ , SiC,  $TiB_2$ ,  $B_4C$ ,  $Si_3N_4$ , SiAlON and PSZ.
7. The invention of claim 4 wherein said pore formers comprise wax.
8. The invention of claim 4 wherein said pore formers comprise  $CaCl_2$ .
9. The invention of claim 4 wherein said pore formers have lubricating properties.
10. The invention of claim 4 wherein the surfaces of said pore formers are coated with oil before being intermixed with said mixture of said curable resin with said sinterable ceramic powder.
11. A method of making a porous ceramic body of high strength and integrity comprising:
  - a.) forming a predominantly homogenous mixture of curable resin and sinterable ceramic powder;

b.) admixing said mixture of said resin and said powder to a plurality of pore formers to form an admixture;

c.) consolidating said admixture such that substantially all voids are eliminated;

d.) curing said resin within said admixture to form a green body;

e.) removing said pore formers from said green body; and

f.) after said pore formers are removed from said green body, firing said green body to sintering temperature and sintering said powder to form said porous ceramic body of high strength and integrity;

wherein said resin and said pore formers are selected such that wettability of said resin in respect to said pore formers is such that said resin tends to bead when in contact with the surfaces of said pore formers before said resin is cured.

12. The invention of claim 11 wherein said sinterable ceramic powder is selected from the group comprising  $\text{Al}_2\text{O}_3$ , SiC,  $\text{TiB}_2$ , BaC,  $\text{Si}_3\text{N}_4$ , SiAlON and PSZ.

13. The invention of claim 11 wherein said curable resin is selected from the group comprising polyester, epoxy, polyethylene, polypropylene, phenolic and polyvinylchloride.

14. The invention of claim 11 wherein said curable resin is mixed with a plasticizer before it is formed into said mixture of said curable resin and said sinterable ceramic powder.

15. The invention of claim 11 wherein said curable resin is mixed with a lubricant before it is formed into said mixture of said curable resin and said sinterable ceramic powder.

16. The invention of claim 11 wherein said curable resin is mixed with a curing agent before it is formed into said mixture of said curable resin and said sinterable ceramic powder.

17. A porous ceramic body of high strength and integrity produced by a method comprising:

a.) forming a substantially homogenous mixture of curable resin and sinterable ceramic powder;

b.) admixing said mixture of said resin and said powder to a plurality of pore formers to form an admixture;

c.) consolidating said admixture such that substantially all voids are eliminated;

d.) curing said resin within said admixture to form a green body;

e.) removing said pore formers from said green body; and

f.) after said pore formers are removed from said green body, firing said green body to sintering temperature and sintering said powder to form said porous ceramic body of high strength and integrity;

wherein said resin and said pore formers are selected such that wettability of said resin in respect to said pore formers is such that said resin tends to bead when in contact with the surfaces of said pore formers before said resin is cured.

18. The invention of claim 17 wherein said sinterable ceramic powder is one of the group comprising  $\text{Al}_2\text{O}_3$ , SiC,  $\text{TiB}_2$ , BaC,  $\text{Si}_3\text{N}_4$ , SiAlON and PSZ.

19. The invention of claim 17 wherein said curable resin is selected from the group comprising polyester, epoxy, polyethylene, polypropylene, phenolic and polyvinylchloride.

20. The invention of claim 17 wherein said curable resin is mixed with a plasticizer before it is formed into said mixture of said curable resin and said sinterable ceramic powder.

21. The invention of claim 17 wherein said curable resin is mixed with a lubricant before it is formed into said mixture of said curable resin and said sinterable ceramic powder.

22. The invention of claim 17 wherein said curable resin is mixed with a curing agent before it is formed into said mixture of said curable resin and said sinterable ceramic powder.

23. A porous ceramic body of high strength and integrity formed by removing the pore formers from, and then sintering, a green body which comprises a consolidation of a predominantly homogenous mixture of a curable resin with a sinterable ceramic powder which is intermixed with a plurality of said pore formers, the materials of said resin and said pore formers being selected such that the wettability of said resin in respect to said pore formers is such that said resin tends to bead when in contact with the surfaces of said pore formers, before said resin is cured, said resin in said consolidation which is cured and said pore formers in said consolidation which are removable after said resin in said green body has been cured.

FIG. 1

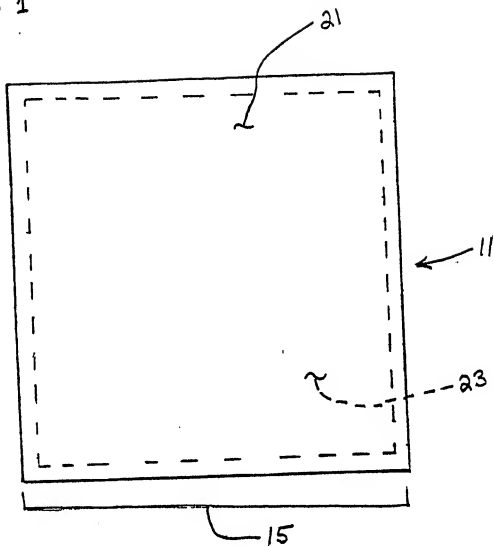
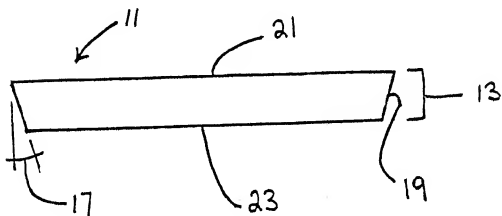


FIG. 2



# FIG. 3

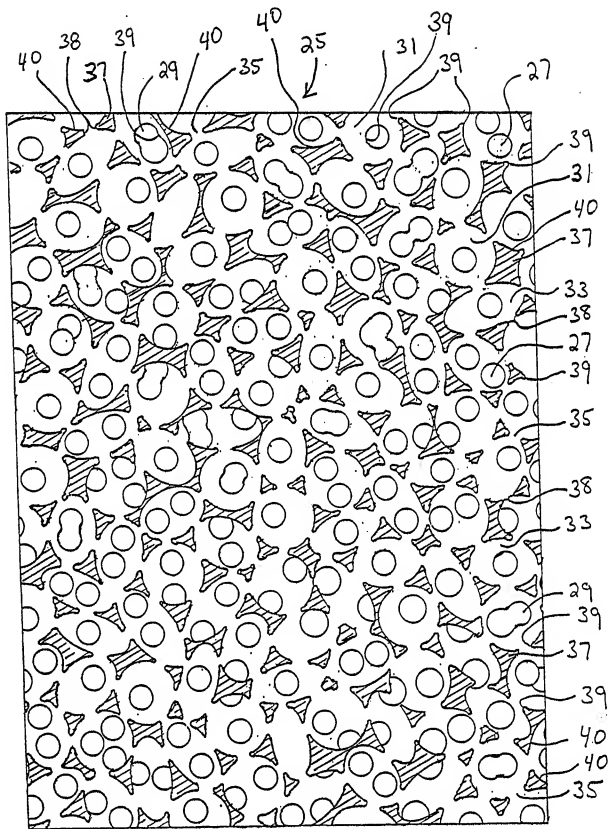


FIG. 4

